Synthesis of Novel Spiro-condensed Dithienosiloles and the Application to Organic FET

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Spirobi(dithienosilole)s were prepared and their optical and FET properties were studied. They showed absorption maxima at 358–368 nm, a little red shifted from those of the corresponding non-spiro type dithienosiloles, indicating that spiro-conjugation operates in these molecules to an extent. The field-effect hole mobility of spirobi[bis(trimethylsilyl)dithienosilole] was determined to be 1.4×10^{-6} cm² V⁻¹ s⁻¹.

A couple of years ago, we have initiated studies concerning the synthesis and properties of dithienosilole derivatives having a silole system condensed with a bithiophene system (DTS).¹ In these molecules, the silicon bridge retains the coplanarity of two thiophene rings to enhance the conjugation and, moreover, interaction between σ^* orbital of the silole silicon and π^* orbital of the inner ring butadiene unit lowers the LUMO energy level, making it possible to use these compounds as electron transporting materials for EL devices.² It was also found that some polymers having DTS units in the backbone would be used as holetransports in EL devices.³ In an effort to explore further the scope of the dithienosilole system, we prepared spiro-condensed ones as the novel σ – π conjugated compounds.



Chart 1. Dithienosilole (DTS)

Spiro-condensed dithienosiloles were readily obtained, as shown in Scheme 1.⁴ Thus, the reaction of 3,3'-dilithio-5,5'-bis-(trimethylsilyl)-2,2'-bithiophene with tetrachlorosilane gave spirobi[2,6-bis(trimethylsilyl)dithienosilole] (sDTS-1), as the pale vellow crystals whose melting point was observed at 308 °C by DSC higher than those of simple DTS derivatives; e.g., DTS-1 (R = SiMe₃, Ar = Ph in Chart 1) melted at 158– 160 °C.² Similar increase of morphological thermal stability of π -conjugated systems by spiro-condensation has been often observed.⁵ Subsequent bromination of sDTS-1 gave sDTS-Br. Lithiation of sDTS-Br with n-butyllithium, followed by treatment with chlorosilanes gave substitution products sDTS-2 and sDTS-3. They melted at 204-205 °C and 223-225 °C, respectively. The present sDTS exhibited UV absorption and emission maxima at 366-368 nm (E 21,000-26,000) and 434-436 nm (Φ 0.10–0.17), respectively, which were red-shifted from those of **DTS-1** by about 10 nm.² Cyclic voltammometric analysis of sDTS-1 in THF/acetonitrile = 1/4 revealed an irreversible oxidation peak at 0.93 V vs Ag/Ag⁺, at almost the same potential as that of **DTS-1**.

To know more about the electronic states of the present sDTS, we carried out molecular orbital (MO) calculations of



Scheme 1. Preparation of sDTS.

the models, at the level of HF/6-31G^{*,6} As shown in Figure 1, both HOMO and LUMO energy levels are affected by spiro-condensation leading to a decrease of the HOMO–LUMO energy gap, presumably by through space interaction between two orthogonal bithiophene π -orbitals, namely spiro-conjugation.⁷ It is also noted that $\sigma^*-\pi^*$ interaction takes place efficiently in the spiro-condensed system, as indicated by the LUMO orbital profile that shows mixing of Si σ^* and thiophene π^* orbitals clearly (Figure 1), similar to **DTS**.



Figure 1. Relative HOMO and LUMO energy levels for model compounds, derived from MO calculations, and HOMO and LUMO orbital profiles for **sDTS**.

The crystal structure of **sDTS-1** was determined by a single crystal X-ray diffraction study and the ORTEP drawing is depicted in Figure 2.⁸ Two dithienosilole rings of **sDTS-1** retain high coplanarity and are located in a perpendicular fashion to each other. The bond distances and angles in DTS systems closely resemble those of **DTS-2** (R = H, Ar = Ph in Chart 1),¹ indicating configuration strain arising from the spiro-condensation is negligible if at all present.

We examined a vapor-deposited film of **sDTS-1** as an FET semiconductor.⁹ Figure 3 shows typical plots of the i_d versus V_d



Figure 2. ORTEP drawing of **sDTS-1**. Thermal ellipsoids are drawn at 50% probability level. Hydrogen atoms are omitted for clarity.

at various V_g for the FET based on **sDTS-1**. The channel conductance increases as V_g becomes more negative, implying that the **sDTS-1** film behaves as a p-channel semiconductor. The field-effect hole mobility μ_{FET} estimated for the **sDTS-1** film was $1.4 \times 10^{-6} \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$. Although the mobility was still low, this may be improved by optimization of the deposition conditions that are known as important factors for the FET performance. **sDTS-1** is the first example of FET active oligothiophene derivatives having π -conjugated units shorter than quauterthiophene.¹⁰ Moreover, **DTS-1** did not show any FET activities, indicating that the spiro-condensation is responsible for the FET properties of **sDTS-1**. Spiro-condensed compounds with orthogonal chromophores have been extensively studied as organic electronic materials.⁵ However, no FET properties of these compounds have not yet been reported.



Figure 3. i_d vs V_d curves at different gate biases for the OFET device using a thin film of **sDTS-1** as the active layer.

In conclusion, we prepared novel spiro-condensed dithienosiloles by a readily accessible process. The elevated melting points by spiro-condensation seem to suggest the improved morphological stability in films. In addition, the preparation and reactions of **sDTS-Br** may allow further transformation of the spiro-compounds, leading to such as polymers and dendrimers. We also demonstrated the potential utilities of these types of the compounds as the FET materials, which may provide a new strategy for the molecular designing.

References and Notes

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- Data for sDTS-1: MS m/z 644 (M⁺); ¹H NMR (δ in CDCl₃) 0.32 (s, 36H), 7.07 (s, 4H); 13 C NMR (δ in CDCl₃) 0.05, 136.91, 137.16, 142.14, 157.24; ²⁹Si NMR (δ in CDCl₃) -6.65, -39.51. Anal. Calcd for C₂₈H₄₀S₄Si₅: C, 52.11; H, 6.25. Found: C, 52.14; H, 6.26%; sDTS-2: TOF-MS m/z 893.18; $^1\mathrm{H}\,\mathrm{NMR}$ (δ in CDCl_3), 0.56 (s, 24H), 7.04 (s, 4H, thiophene); 7.33–7.38 (m, 12H, *m*- and *p*-Ph), 7.56 (dd, 8H, J =7.5 and 1.9 Hz, *o*-Ph); ¹³C NMR (δ in CDCl₃) –1.31, 127.90, 129.43, 133.90, 136.96, 137.54, 138.37, 140.04, 157.76; ²⁹Si NMR (δ in CDCl₃) -11.73, -39.37. Anal. Calcd for C₄₈H₄₈S₄Si₅: C, 64.52; H, 5.41. Found: C, 64.33; H, 5.46%; **sDTS-3**: MS m/z 588 (M⁺); ¹HNMR (δ in CDCl₃) 0.37 (d, 24H, J = 3.6 Hz), 4.53 (septet, 4H, J = 3.6 Hz, -SiH), 7.09 (s, 4H, thiophene); 13 C NMR (δ in CDCl₃) -2.82, 136.96, 138.17, 138.58, 157.69. Anal. Calcd for C24H32S4Si5: C, 48.92; H, 5.47. Found: C, 49.01; H, 5.49%.
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- Crystal data for sDTS-1: space group $I4_1/a$ (#88) with 8 a = 24.5701(3) Å, c = 24.7008(3) Å, V = 14911.6(3) Å³, Z = 16, $D_{calcd} = 1.150 \text{ g/cm}^3$, F(000) = 5471.00, μ (Mo $K\alpha$) = 4.32 cm⁻¹. 8507 Intensity data were collected at 108 K on a Rigaku RAXIS-RAPID imaging plate diffractometer using a $0.5 \times 0.3 \times 0.2 \text{ mm}^3$ sized crystal. 5776 unique reflections; with $I > 3^{\circ}(I)$ (reflns/para = 15.44) were used in refinement; R = 5.4, Rw = 8.7%. Selected bond distances (Å) and angles (deg): Si(1)-C(2) = 1.875(6), Si(1)-C(6) =1.869(9), Si(1)-C(10) = 1.876(6), Si(1)-C(14) = 1.865(6), S(1)-C(1) = 1.717(6), S(1)-C(4) = 1.741(6), C(1)-C(2) =1.378(8), C(1)-C(5) = 1.462(8), C(2)-C(3) = 1.419(8),C(5)-C(6) = 1.385(8), C(2)-Si(1)-C(6) = 92.2(3), C(2)-C(6) = 92.2(3), C(2)-C(6) = 92.2(3), C(2)-C(6) = 0.2(3), C(2), C(2),Si(1)-C(10) = 119.6(3), C(6)-Si(1)-C(10) = 119.3(3), C(2)-C(10) =Si(1)-C(14) = 116.8(3), C(6)-Si(1)-C(14) = 119.1(3),C(10)-Si(1)-C(14) = 92.3(3). Four thiophene and two silole rings in the molecule show essentially the same structural parameters, respectively. Crystallographic data for sDTS-1 have been deposited with Cambridge Crystallographic Data Centre as supplementary publication no. CCDC-239560.
- 9 An FET based on **sDTS-1** was fabricated on doped Si wafers with a 230 nm thermally grown SiO₂ as a bottom contact type. The drain-source channel length and width are $5 \mu m$ and 20 mm, respectively. Field-effect characteristics of the device were measured under vacuum at room temperature. The field-effect mobility was calculated in the saturation regime of the i_d as shown in a previous paper. See, Y. Kunugi, K. Takimiya, K. Yamane, K. Yamashita, Y. Aso, and T. Otsubo, *Chem. Mater.*, **15**, 6 (2003). XRD measurement of the vapordeposited film showed no signals, indicating that the film is amorphous.
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